

Research Article

Effect of Particle Size and Lattice Strain on the Debye-Waller Factors of Copper (Cu) Powder Using High Energied Ball Mill

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Abstract

Lattice strains in Cu powder produced by milling have been analyzed by X-ray powder diffraction. The lattice strain (ϵ) and Debye-Waller factor (B) are determined from the half-widths and integrated intensities of the Bragg reflections. In Cu, the Debye-Waller factor is found to increase with the lattice strain. From the correlation between the strain and effective Debye-Waller factor, the Debye-Waller factors for zero strain have been estimated for Cu. The variation of energy of vacancy formation as a function of lattice strain has been studied.

Keywords: X-ray diffraction, lattice strain, crystallite size, Debye-Waller factor, vacancy formation energy.

1. Introduction

Copper is one of the most widely used structural ceramics [1,2]. Several properties of alumina are controlled by its microstructure and its matrix stability. The Debye-Waller factor is an important lattice dynamical property. There is considerable X-ray work on the Debye-Waller factors of Cu [3-4]. But it is interesting to study the effect of lattice strains on the Debye-Waller factors of these metals. Inagaki *et al* [5,6] showed that in several non-metallic powders, the strains produced during grinding have a significant effect on the Debye-Waller factors measured from X-ray diffraction intensities. Sirdeshmukh *et al* [7] observed the effect of lattice strains on the Debye-Waller factors in semiconductor powder materials. Gopi Krishna and Sirdeshmukh [8] studied the effect of lattice strains on the Debye-Waller factor of ytterbium metal. In the present investigation the results of a systematic study of the effect of lattice strains on the Debye-Waller factors and crystallite size of face centred cubic Cu metal is reported. These results are being reported for the first time.

2. Experimental

Pure copper metal supplied by Fluka (UK) was used for the study. The powder samples were obtained by gently filing highly pure Cu metal ingots with a jeweller's file. A part of this powder was used to prepare the initial sample. The remaining powder was subjected to milling in a ball mill for 4, 8, 12, 16 and 20 hours to produce strains. X-ray diffractograms were recorded with the initial sample and with samples prepared after each spell of milling. The XRD pattern of Cu is given in Fig. 1. The diffractograms were obtained with a Philips CWU 3710 X-ray powder

diffractometer in the 2θ range $20-120^\circ$ using filtered CuK_α at a goniometer speed of 0.5 per minute and a chart speed of 20 mm/min. All measurements were made at room temperature. The observed integrated intensities have been corrected for thermal diffuse scattering using the method of Chipman and Paskin [9].

1. Analysis of Data

For the relative intensity method, the expression for the observed intensities I_0 is given by

$$I_0 = CL_p J F_T^2 \quad (1)$$

where L_p is the Lorentz-polarization factor, J , the multiplicity factor, F_T the structure factor and C is a constant. For a flat powder specimen, the absorption correction is independent of the angle θ , Klug and Alexander [10] and, hence, is lumped with the constant. The structure factor F_T in terms of the structure factor F for the static lattice is given by

$$F_T = F e^{-M} \quad (2)$$

We may also write Eq. (1) as

$$I_0 = I_c e^{-2B \left(\frac{\sin \theta}{\lambda} \right)^2} \quad (3)$$

where I_c is the intensity corresponding to the static lattice and is given by

$$I_c = L_p J F^2 \quad (4)$$

For crystals with fcc structure, the structure factor F is given by

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$$F = 4f \tag{5}$$

$$B_r \cos \theta = \frac{k\lambda}{t} + \epsilon \sin \theta \tag{9}$$

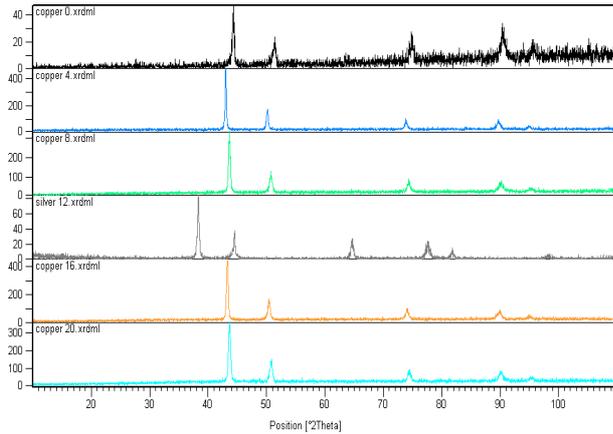


Fig. 1. XRD pattern of Cu

f being the atomic scattering factor. Values of the atomic scattering factor were taken from Cromer and Waber [11] and International Tables for X-ray Crystallography [12], and have been corrected for dispersion, Cromer and Liberman [13]. The choice of the atomic scattering factors was guided by Cromer’s [11] recommendation who suggested that the atomic scattering factors given in the International Tables for X-ray Crystallography [12] which are computed from the Hartree-Fock wave functions are the best for the lighter atoms or ions in the periodic table up to Rb^{+1} . For the heavier atoms or ions Cromer recommends the use of the Dirac-slater relativistic scattering factors given by Cromer and Waber [11].

From Eq. (3) it can be seen that $\log(I_0/I_c)$ is linearly related to $(\sin \theta / \lambda)^2$. By a least square treatment of data, B was determined. From the Debye-Waller theory

$$B = \left(\frac{8\pi^2}{3} \right) \langle u^2 \rangle \tag{6}$$

for a cubic crystal, where $\langle u^2 \rangle$ is the mean-square amplitude of vibration. Further, B, may also be expressed as

$$B = \left(\frac{6h^2}{mk_B T} \right) W(x) \tag{7}$$

where m is the mass, T the absolute temperature and h and k_B are the Planck and the Boltzmann constants respectively. The function W(x) is given by

$$W(x) = \left[\frac{\phi(x)}{x^2} + \frac{1}{4x} \right] \tag{8}$$

where $\phi(x)$ is the Debye function and $x = \theta_M/T$, θ_M being the Debye temperature. Benson and Gill [14] have tabulated values of W(x) for a wide range of x for small increments, from which θ_M can be obtained from the value of B.

The total peak broadening B_r may be expressed as,

The plot of $B_r \cos \theta / \lambda$ versus $\sin \theta / \lambda$ is a straight line with slope equal to ϵ and hence the particle size ‘t’ can be estimated from the intercept. Typical Hall-Williamson plot between $B_r \cos \theta / \lambda$ and $\sin \theta / \lambda$ is shown in Figure 2.

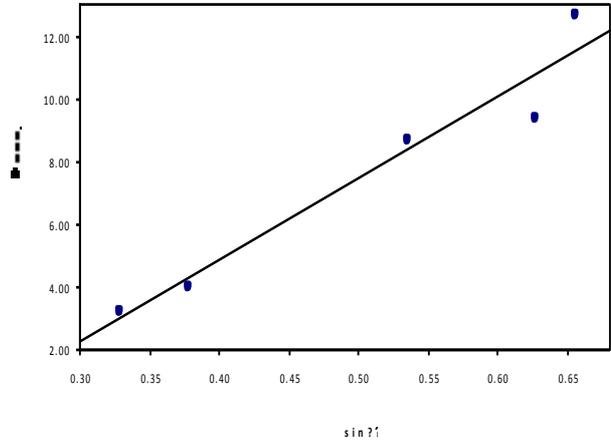


Fig. 2. Plot of $B_r \cos \theta / \lambda$ Vs $\sin \theta / \lambda$ for Ag after milling for 12

The lattice strains were determined from the plot of $B_r \cos \theta / \lambda$ against $\sin \theta / \lambda$ following standard procedures [15]. The measured half-widths were corrected for instrumental broadening with reference to a pure strain-free silicon powder. The variation of particle size with milling time is within the limits of experimental errors. This shows that while the milling is enough to create strains, it affect the particle size to a measurable extent. A typical Hall-Williamson plot is shown in in Figure 2.

4 Results and discussion

The values of the lattice strain, crystallite size, root mean square amplitude of vibrations, Debye-Waller factor and Debye temperature of Cu powder, ground for different durations, obtained in the present study are given in Table 1. As the objective of the present work is to investigate the strain dependence of Debye-Waller factors, the variation of the lattice strain (ϵ) and Debye-Waller factor (B) for different milling times the fcc metal Cu is shown in Figure 3. Both lattice strain and Debye-Waller factor increase with milling time. This is similar to the observations of Inagaki et al [5,6], Sirdeshmukh et al [7] and Gopi Krishna and Sirdeshmukh [8]. The Debye-Waller factor increases with milling time and lattice strain in a slightly non-linear fashion. Crystallite size decreases with milling time. An extrapolation of the B versus ϵ curve to $\epsilon = 0$ gives the values of Debye-Waller factor 0.56 \AA^2 for Cu. The zero-strain value of Debye-Waller factor is less than the value for the initial sample. The zero strain Debye-Waller factors of 0.56 \AA^2 for Cu is close to the value of 0.55 \AA^2 obtained by Flinn and Mc Manus [2], using single crystal X-ray diffraction. Thus, the Debye-Waller factor of Cu powder sample carry an effect due to lattice strain. While comparing the Debye-Waller factor calculated from the lattice dynamical model with experimental results, Vetelino *et al* [16] have attributed the difference to inaccuracies in the experimental values caused by neglecting the TDS

corrections. The repeated milling of the powder sample leads to lattice distortion which gives rise to microstrains in the lattice. These microstrains increase the contribution of static component of Debye-Waller factor. Thus both lattice strain and the observed Debye-Waller factor, which is the sum of static and thermal components, increase with milling time. Thus, whenever Debye-Waller factors are determined from X-ray intensities on powder samples, it is desirable to make an estimate of the lattice strain and if the strain is large, a suitable correction is to be made as done in the present study. The Debye temperature derived from the zero strain value of Debye-Waller factor of Cu is 315K. This value agree well with the values of 315K [3] obtained for Cu from the single crystal X-ray diffraction.

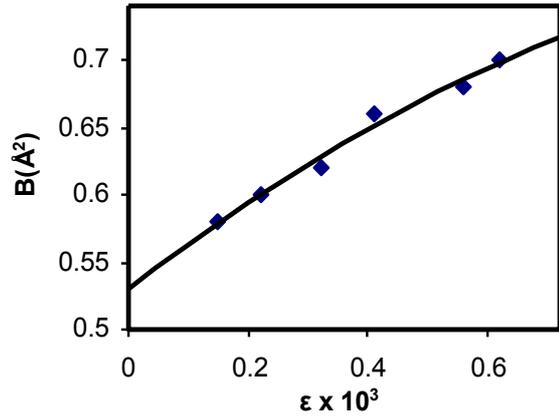
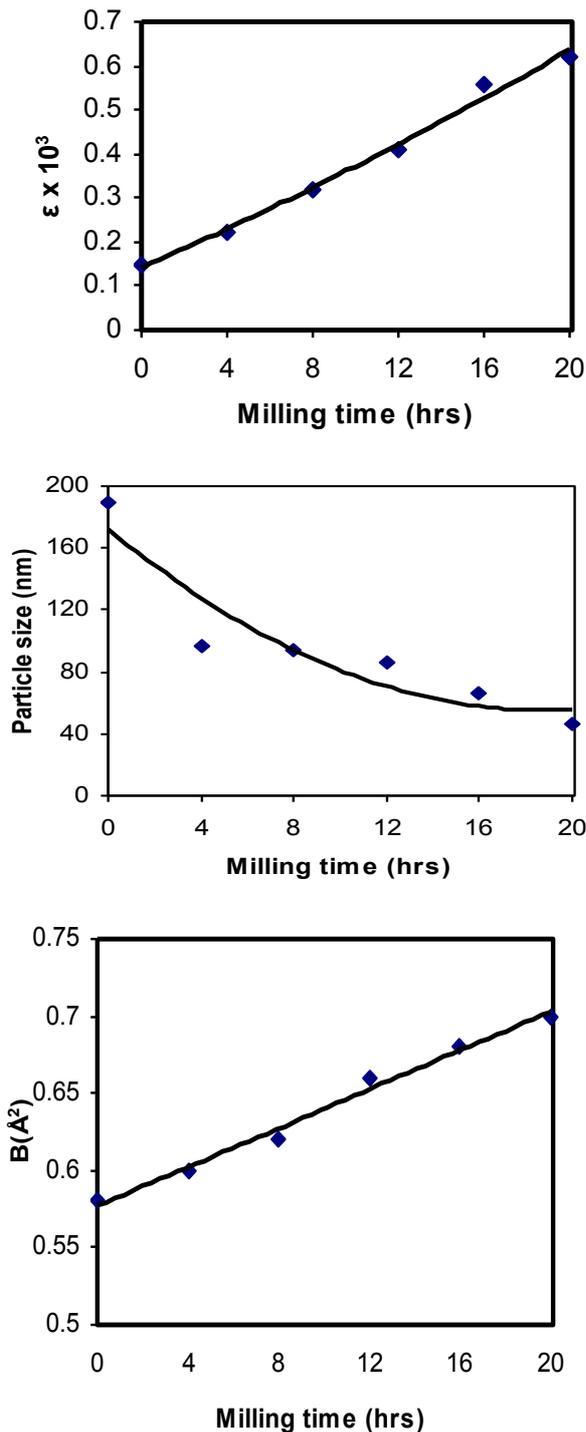


Fig. 3. Milling time *V*s lattice strain, Crystallite size and Debye-Waller factor and lattice strain *V*s Debye-Waller factor curves for Al, Cu and Ag.

Table 1. Values of lattice strain (ϵ), crystallite size (t), Debye-Waller factor (B), root mean square amplitudes of vibration $\langle u \rangle$, Debye temperature (θ_M) and energy of vacancy formation (E_f) of strained Al, Cu and Ag powders.

Metal	Milling time	$\epsilon \times 10^3$	$t(\text{nm})$	$\langle u \rangle(\text{\AA})$	$B(\text{\AA}^2)$	$\theta_M(\text{K})$	$E_f(\text{eV})$
Cu	0	0.15	189	0.1496 (6)	0.58 (3)	309 (23)	1.66
	4	0.22	97	0.1509 (7)	0.60 (4)	306 (31)	1.63
	8	0.32	94	0.1535 (4)	0.62 (1)	299 (10)	1.56
	12	0.41	86	0.1584 (6)	0.66 (3)	290 (17)	1.46
	16	0.56	66	0.1607 (6)	0.68 (3)	285 (17)	1.41
	20	0.62	46	0.1631 (5)	0.70 (2)	281 (9)	1.37

Glyde [17] derived the following relation between the energy of vacancy formation (E_f) and the Debye temperature (θ) of a solid. The relation is

$$E_f = A(k/h)^2 M\theta^2 a^2 \tag{5}$$

where a is the interatomic spacing, A a constant shown to be equal to 1.17×10^{-2} , M the molecular weight and h and k are the Planck's and the Boltzmann's constants, respectively. Glyde recommended the use of X-ray based values in eq. (5). The validity of eq.(5) was verified for a number of fcc, bcc and hcp metals [18]. Therefore, the X-ray Debye temperatures obtained in the present work have been used to study the variation of vacancy formation energy as a function of lattice strain in Cu. The values of vacancy formation energies are also included in Table 1.

5 Conclusion

Cu powder was strained by milling for 20 hours. From a study of X-ray diffractograms recorded at different stages of milling, it is observed that milling for 20 hours has systematic effect on the particle size. However, the milling produces lattice strain and also enhances the effective Debye-Waller factor. By an extrapolation of the plot between the Debye-Waller factor and the lattice strain, the zero strain Debye-Waller factor is obtained for Cu. The variation of energy of vacancy formation as a function of lattice strain has been studied.

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